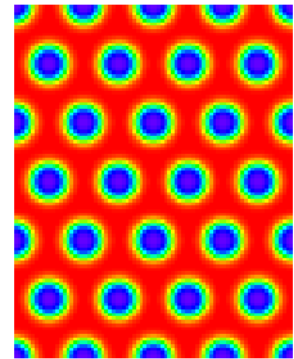
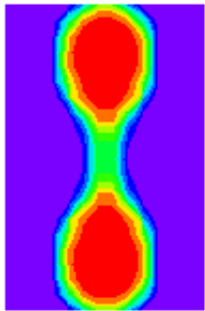


# Self Assembly of Charged Polymers

Galen T. Pickett

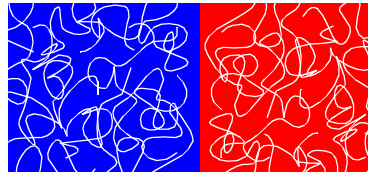
Department of Physics and Astronomy,  
California State University Long Beach



● **Stuff we want to do:**

---

□ **Strengthen mixtures of plastics**



- ☞ Incompatible plastics
- ☞ Combine properties (strength, flexibility)

□ **Lubricate/protect surfaces**

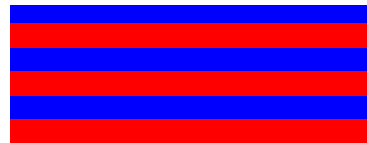
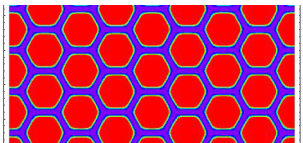


- ☞ Prevent contact
- ☞ Avoid damage

□ **Encapsulate drugs**



□ **Create patterns**

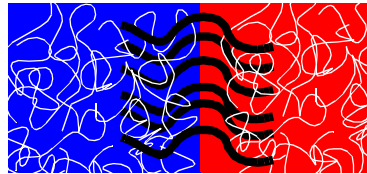


- ☞ Symmetry, scale

● **Stuff that can do it.**

---

□ **Stitching polymers: reinforce mixtures**



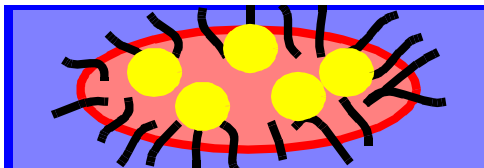
→ Half blue/half red reinforces interface.

□ **End-grafted polymers: lubrication**



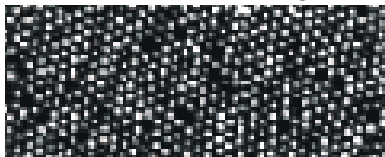
→ Trapped coating  
→ "Osmotic" barrier

□ **Amphiphilic polymers: housing for droplets**



→ Polymer forms vesicles  
→ Release contents, pH *e.g.*

□ **Block copolymers: templates for ordering**

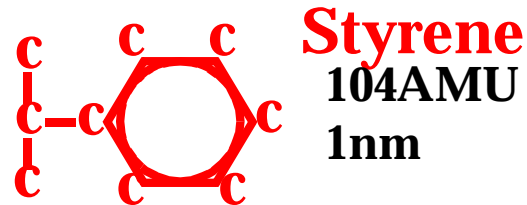


<http://www.princeton.edu/~polymer/>

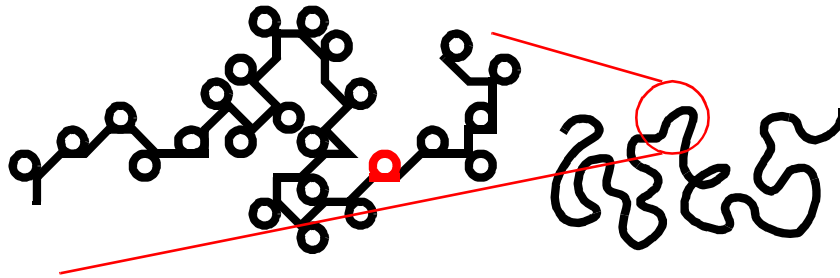
# ● Polymers

---

- Are made of monomers...



- ... strung together into huge chains...



## **Poly**styrene

1000 monomers: 104,000AMU  
1000nm = 1μm

- ... which mostly ignore  $h$ ...

$$\Delta x \Delta p \approx (1 \text{ nm})(10^5 \text{ AMU } v) \approx 10^8 \frac{v}{\text{m/s}} h$$

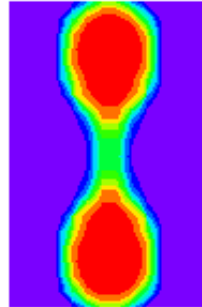
- ... and are all tangled up.



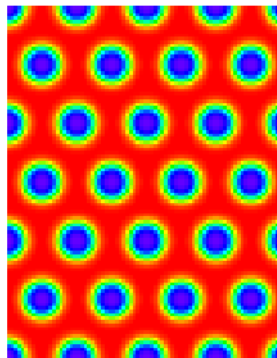
● **Outline:**

---

- **Single-chain self-assembly “folding”**

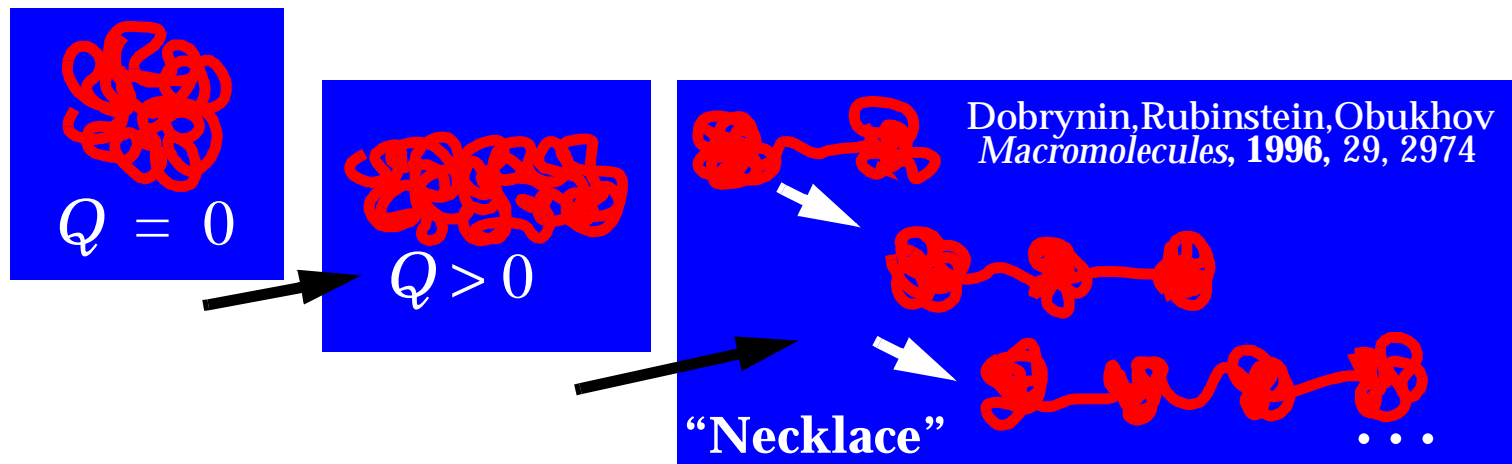


- **Many-chain “super-structures”**



## ● Polyelectrolyte, poor solvent

- Fixed charge  $Q = \alpha N$  on a flexible polymer,  $N$  monomers.
- Poor solvent:
  - ↳  $Q, N$  control conformation.

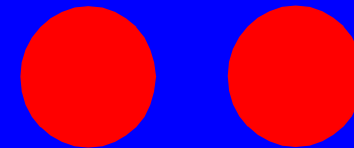
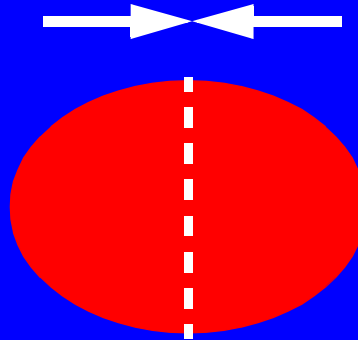
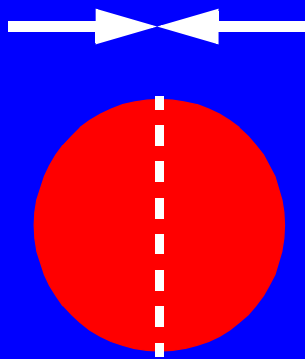


- Cascade of transitions.

## ● **Instability for Charged Oil Drop**

### □ **Lord Rayleigh:**

**Cohesion, surface tension**



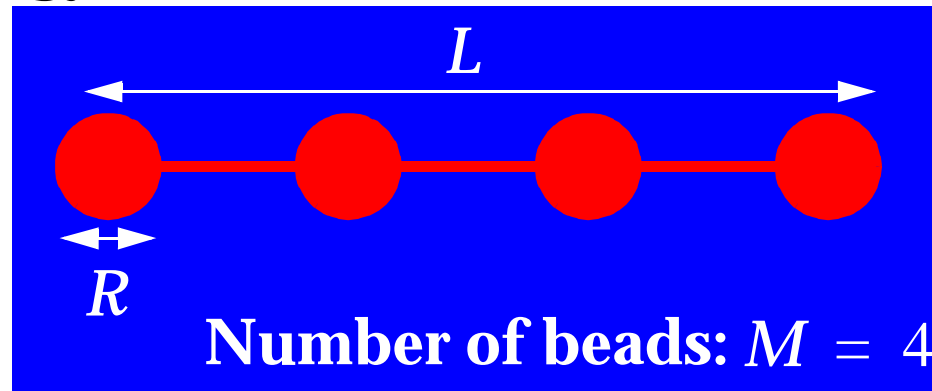
**Electrostatic Self-Interaction**



- **If charge exceeds threshold, surface tension can't maintain the droplet.**
- **Chain *CAN'T* break up.**

- **Cartoon theory for *chain***

- **Free energy of necklace conformation:**



$$F = M \left( \gamma R^2 + \left( \frac{Q}{M} \right)^2 R^{-1} \right) +$$
$$L\gamma + \frac{Q^2}{L}$$

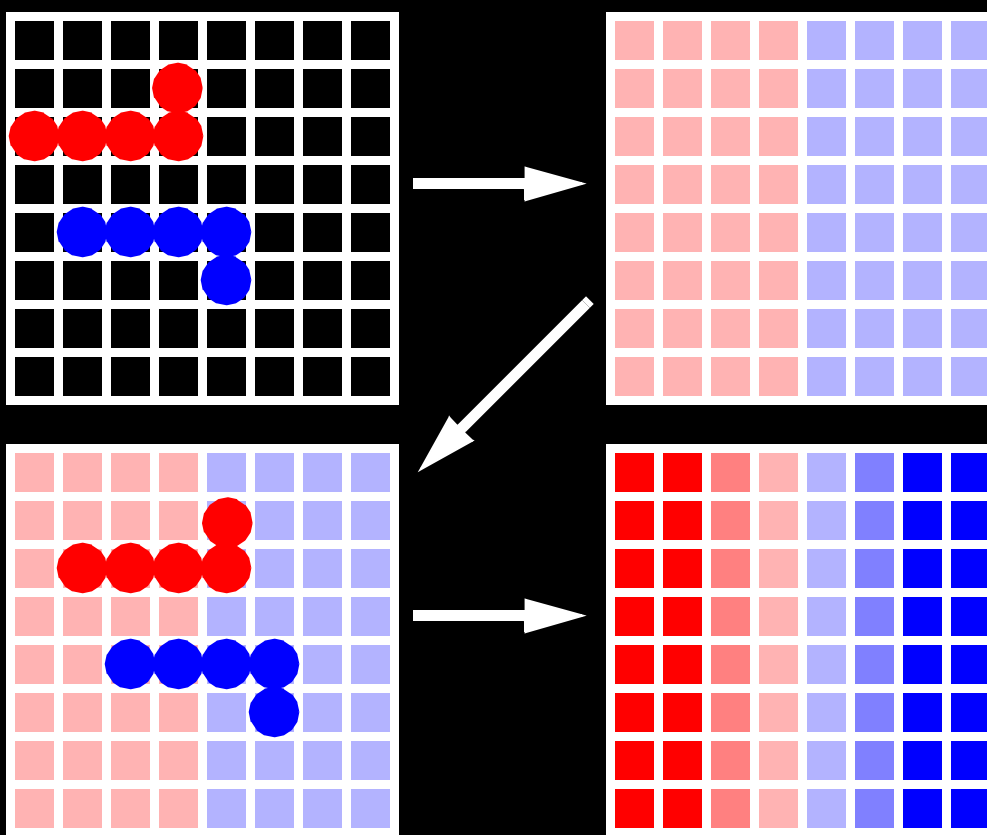
- **Predicts transitions from 1 to 2 to ...**



# ● Self-consistent Lattice Model

Fleer, Cohen, Scheutjens, Cosgrove, Vincent, *Polymers at Interfaces* Chapman and Hall, London 1993

## □ Lattice model



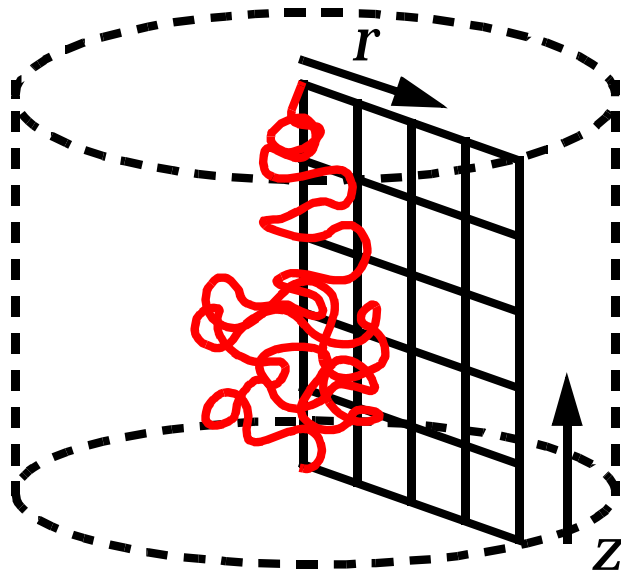
1. Start with empty lattice
2. Throw down polymers at random
3. Calculate average monomer densities
4. Regrow the chains
5. Recalculate the monomer densities
6. Repeat

## □ Azimuthal symmetry, *cylindrical* lattice.

## ● 2-D Cylindrical lattice

---

### □ Azimuthal symmetry:



→  $r, z$  label annular section of three dimensional space.

→ Polymer is held at center of top and bottom plate “bridging”

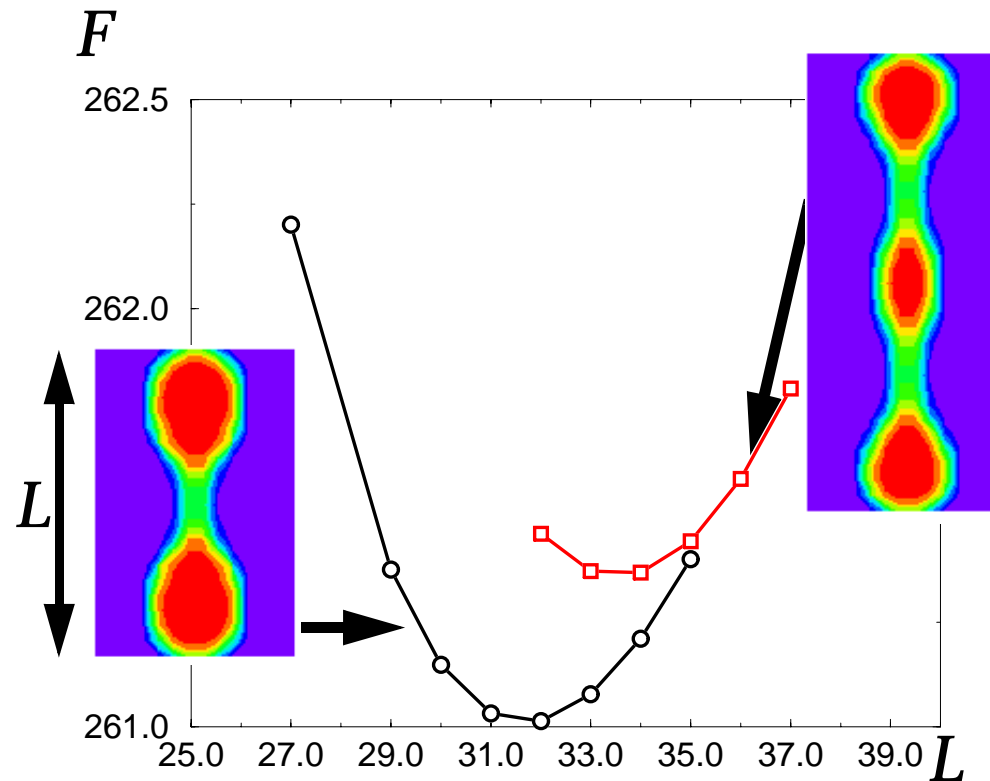
→ Electrostatics, surface energy, chain connectivity are all accounted for

### □ Variations in 2D, but real 3D structures (highly symmetric).

- **Compare structures:**

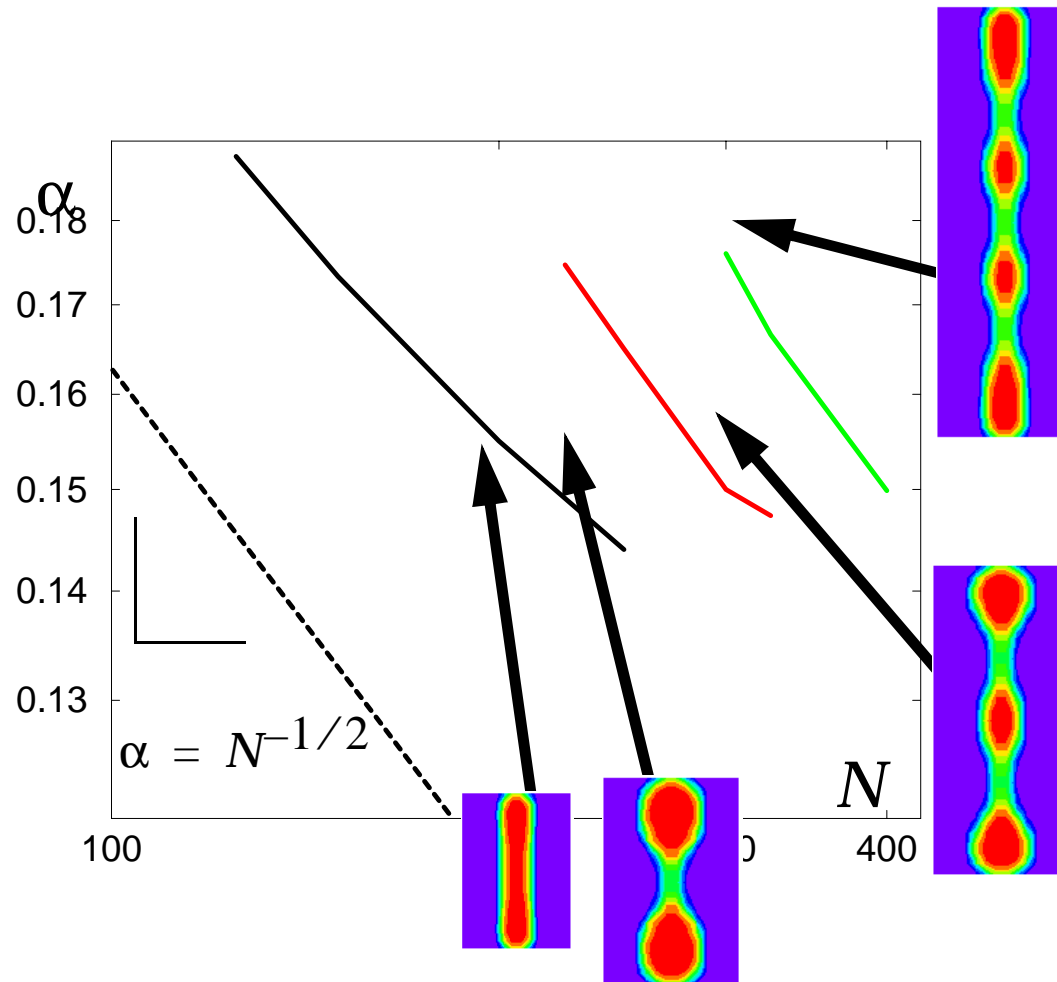
- **At fixed  $N$ ,  $\alpha$  vary  $L$  to find equilibrium structures.**

- ☛  $N = 250, \alpha = 0.16, \chi = 2.0:$



- ☛  $L_{eq}$  minimizes  $F$ . **Possible experiment.**

● **Diagram of states:**

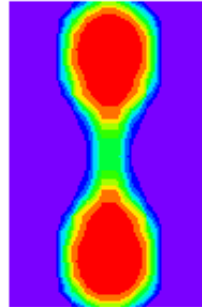


□ **“Folded” conformations.**

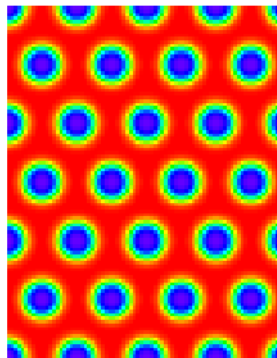
● **Outline:**

---

- **Single-chain self-assembly “folding”**



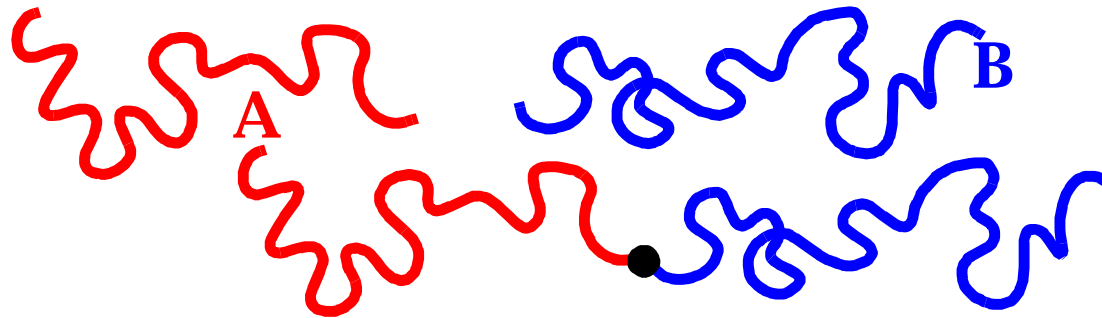
- **Many-chain “super-structures”**



## ● Block copolymers

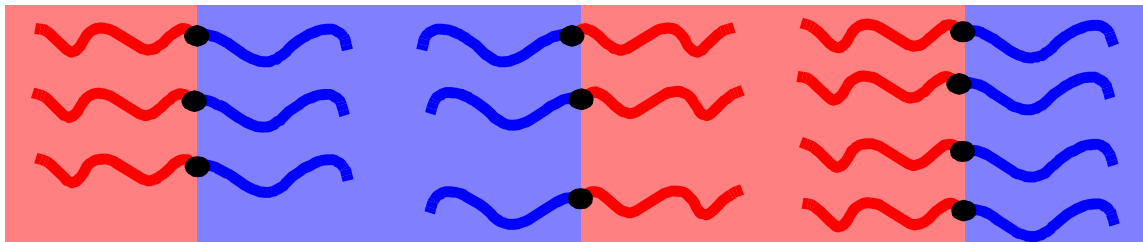
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- Two kinds of monomers strung together.



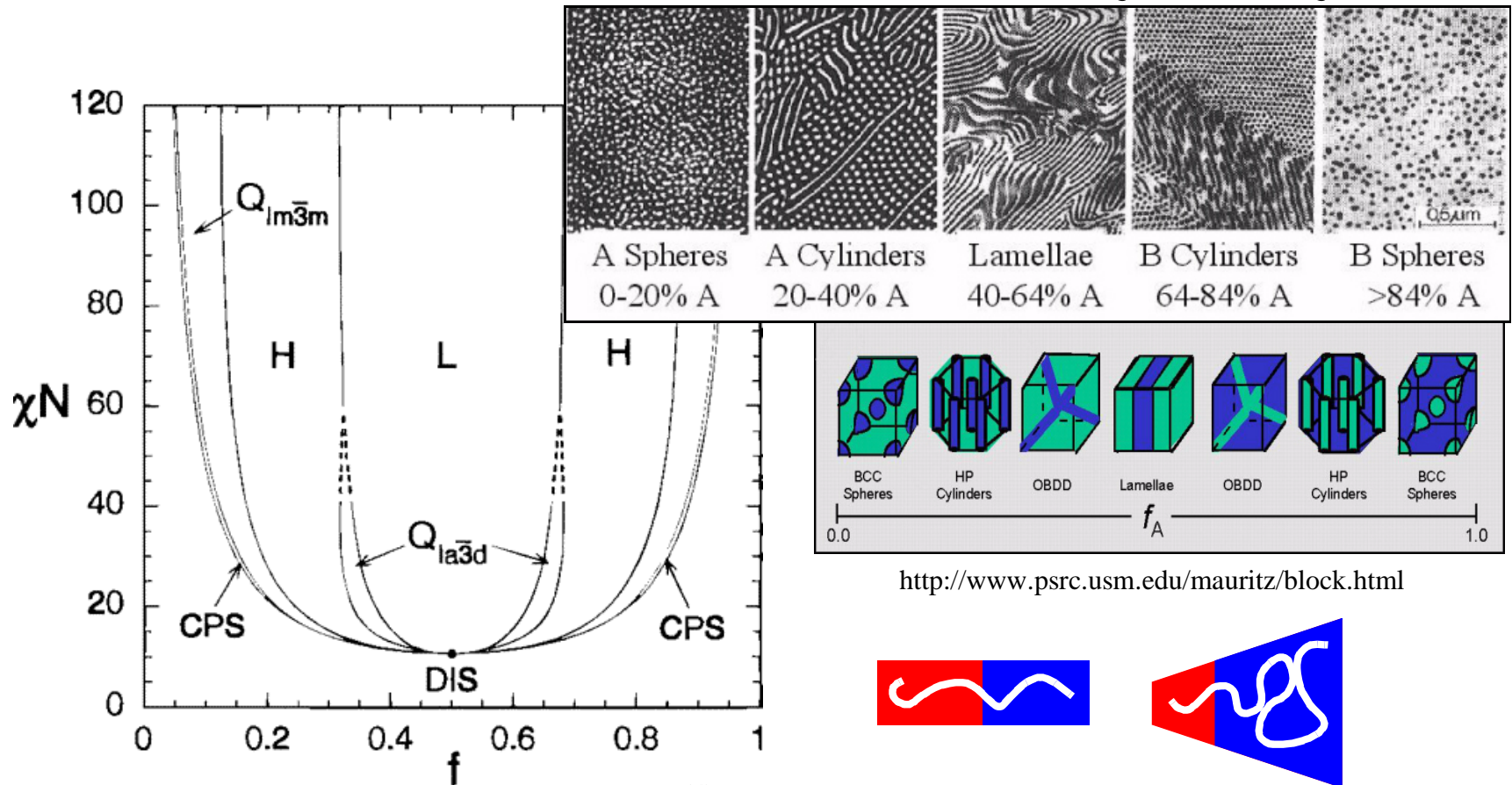
← A-block and B-block: “diblock”

- Unless you break bonds, micro-scale texture happens.



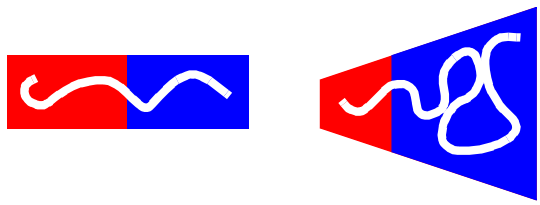
# ● Asymmetric diblocks

□  $f$  = fraction of A on molecule, controls symmetry:



M. W. Matsen and F. S. Bates, *Macromolecules*; 1996; **29**(4); 1091.

<http://www.psrc.usm.edu/mauritz/block.html>



## ● Coarse-grained Free Energy for Diblocks

---

- **Local interactions for order parameter  $\varphi = \varphi_A - \varphi_B$**

$$F_{\text{local}}[\varphi] = \int \left[ \frac{t}{2} \varphi^2 + \frac{k}{2} \nabla \varphi \cdot \nabla \varphi + \varphi^4 \right] dx$$

- **Long-ranged interactions**

$$F_{\text{long-range}}[\varphi] = \int dx \int dx' B \varphi(\mathbf{x}) G(\mathbf{x}, \mathbf{x}') \varphi(\mathbf{x}')$$

- ↳ Ohta and Kawasaki:

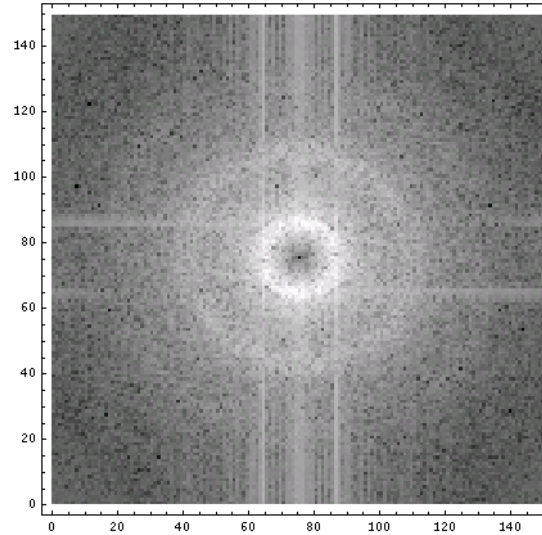
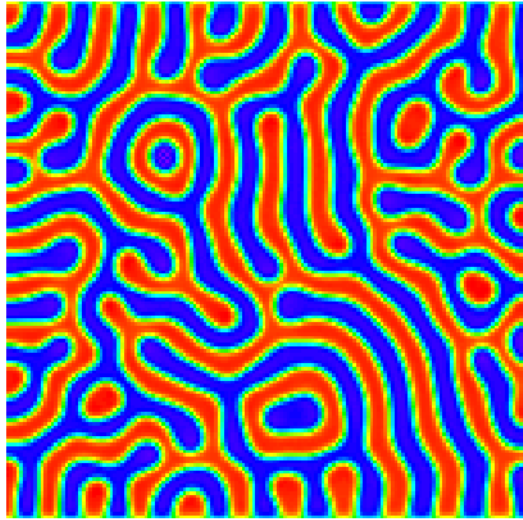
$$\nabla_{\mathbf{x}}^2 G(\mathbf{x}, \mathbf{x}') = -\delta(\mathbf{x} - \mathbf{x}')$$

- **Formally, same as electrostatics.**

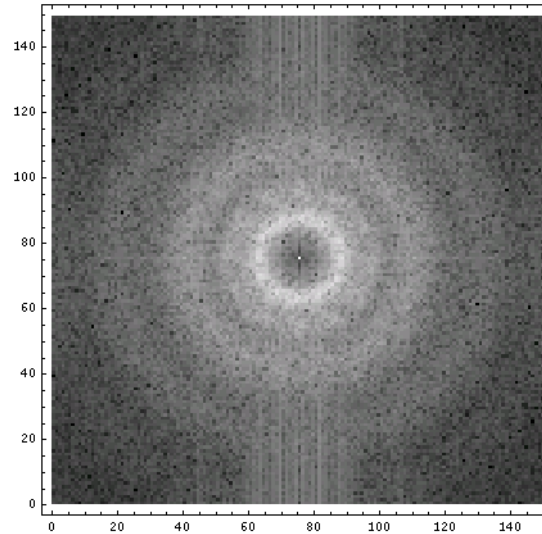
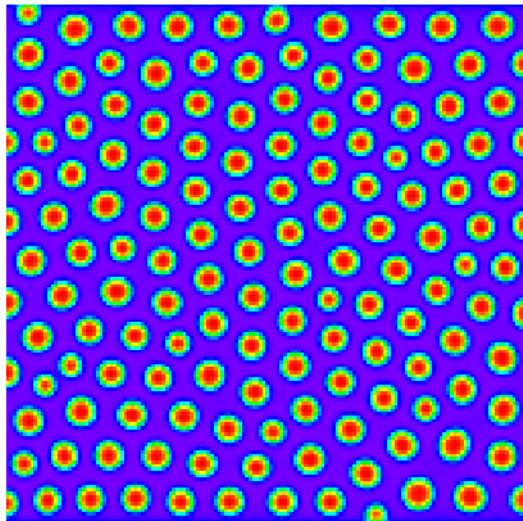
- ↳ **A monomers negative, B monomers positive**



# ● Minimizing $F$ gives diblock-like structures



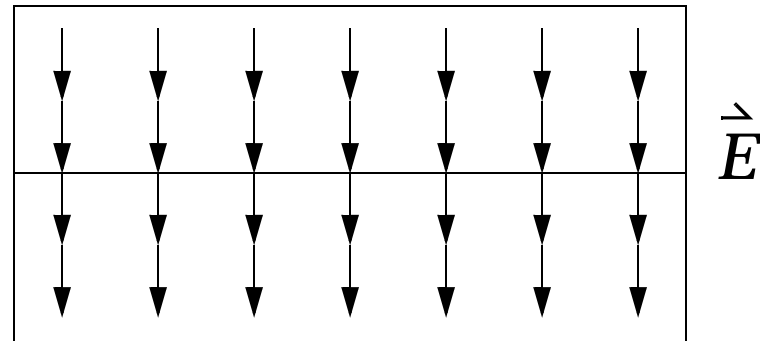
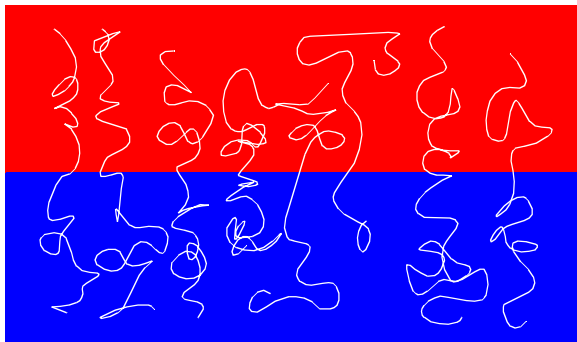
- Cahn-Hilliard dynamics
- Lamellar phase
- Scattering



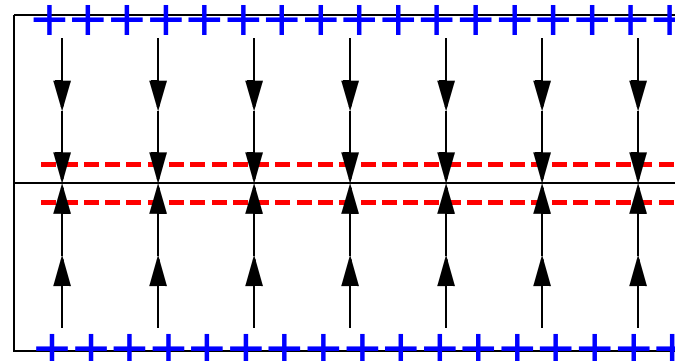
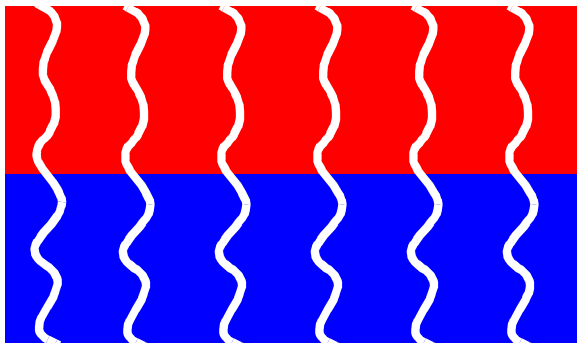
- Cylinder phase
- Scattering

## ● **Electrostatic analogy for Diblocks**

- **Elastic energy**  $\Leftrightarrow$  **Electrostatic self-energy**
- **Semenov, chain stretching similar to electric field:**

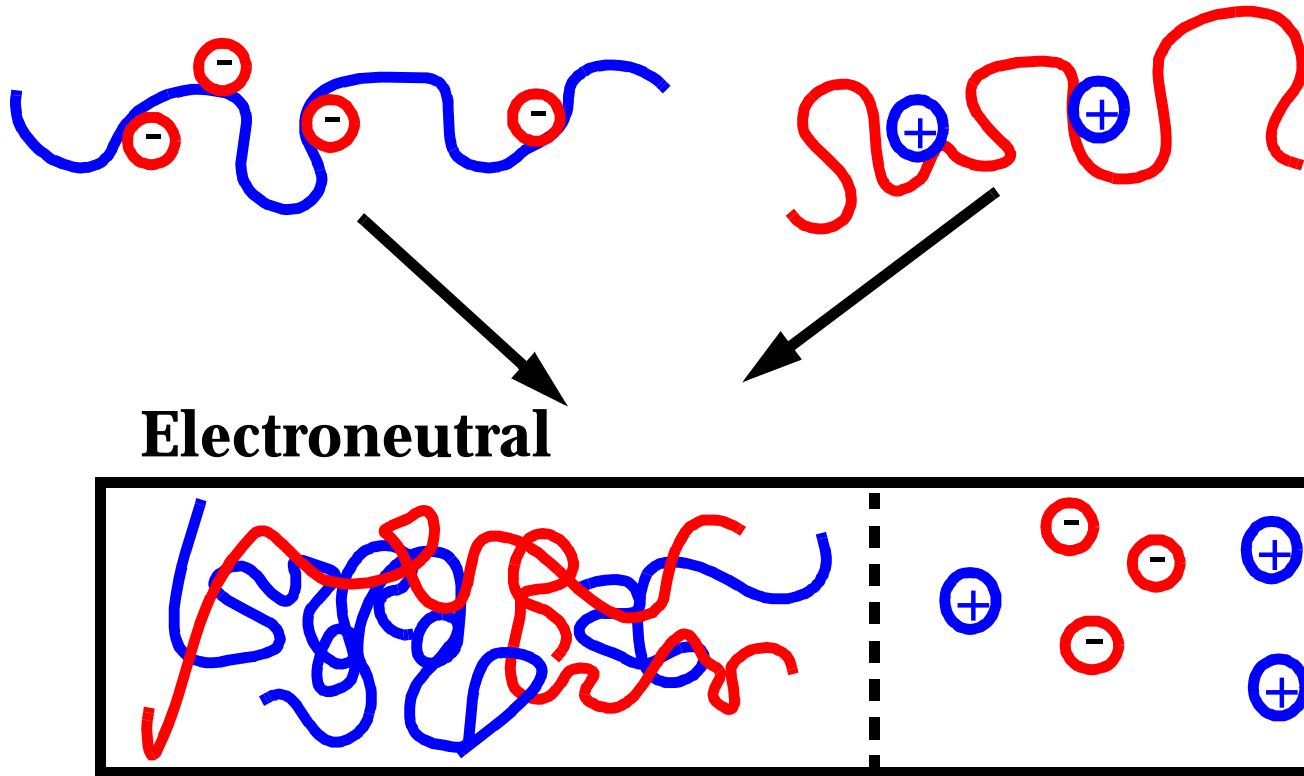


- **Alexander, deGennes, and elaborations**



- **Blend of polyelectrolytes:**

- **Polycation and polyanions mixed together:**



- **Poly-salt melt... what might it do. Phase separate?**

## ● **Blend to consider**

---

- **Let both chains have the same number of monomers (can be relaxed...)**
- **Let the CHARGE/monomer on the majority component be fixed.**
- **Electroneutrality then relates the CHARGE/monomer of minority component to composition:**

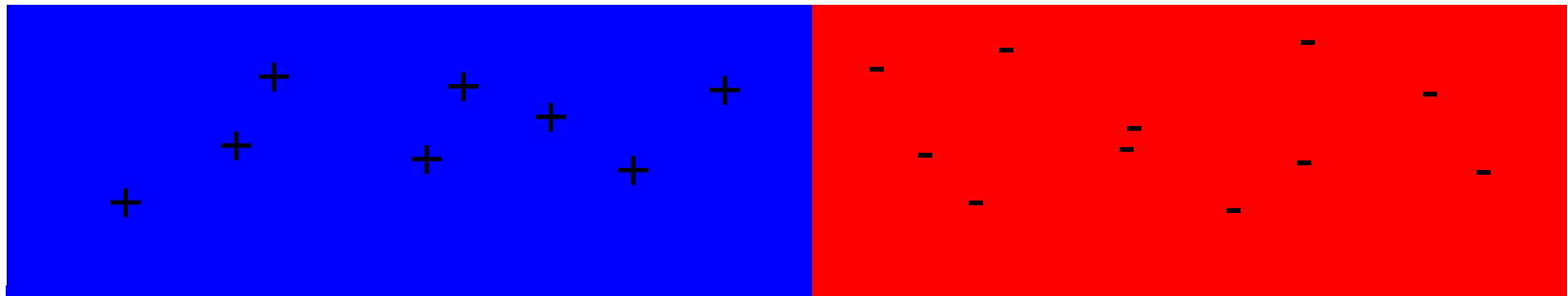
$$0 = \rho_A f + \rho_B (1 - f)$$

- **Minority chain is more strongly charged than majority chain... synthetic chemistry.**

- **Can expect a mesophase.**

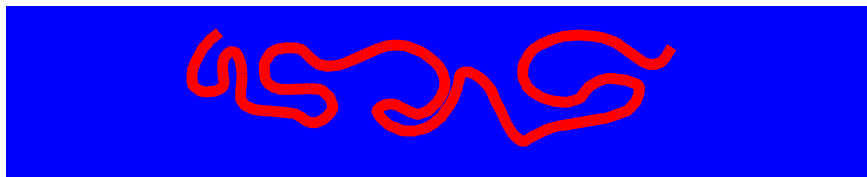
---

- **Phase separation: huge electrostatic costs**



**“Collecting like charges”**

- **Single phase: huge specific interactions**



**N red monomers:  
total cost  $\chi N$**

## ● RPA for Disordered Phase

---

- **Locate linear instability of the uniform phase.**

$$(\varphi_A + \varphi_B = 1) \Rightarrow \varphi_A = f + \varphi, \varphi_B = (1 - f) + \varphi$$

$f$  = fraction of A monomers in system

- **Collective scattering function for order parameter**

$$S^{-1}[q] = S_A^{-1} + S_B^{-1} + \alpha$$

$$\alpha = \frac{\delta^2 F}{\delta \varphi^2} \text{ giving thermal response for fluctuations of } \varphi$$

$$F[\varphi] = \chi \varphi(1 - \varphi) + \frac{2\pi B}{q^2 (1 - f)^2} \varphi^2$$

- **Enforces incompressibility, chain architecture**

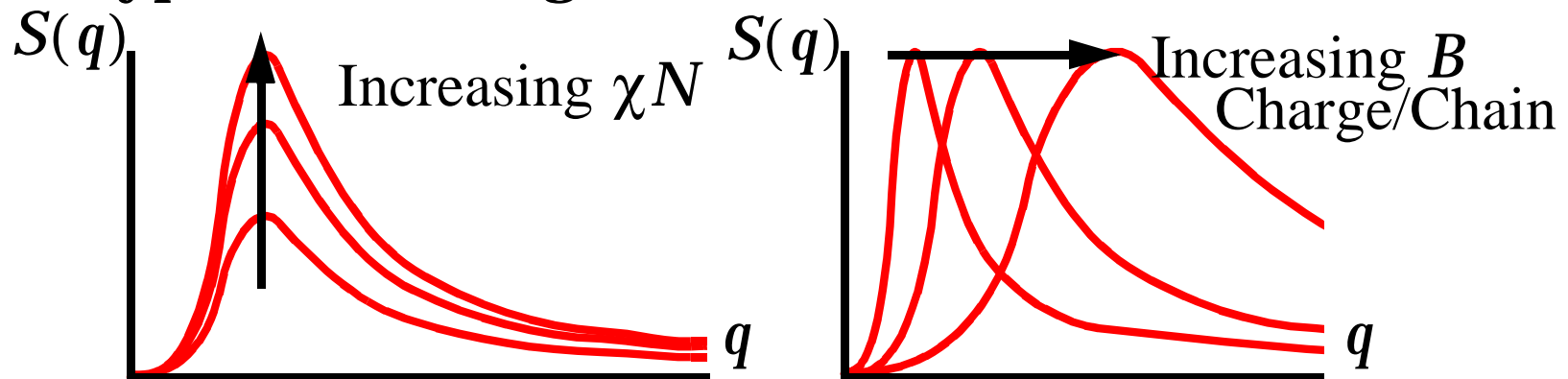
● **RPA cont.**

□ **Single-chain scattering functions:**

$S_A = fg_d(q, N)$  Debye scattering from Gaussian chain

$$g_d(q, N) \approx \frac{N}{1 + Nq^2/12} \quad \text{Lorentzian approximation}$$

□ **Typical scattering function:**

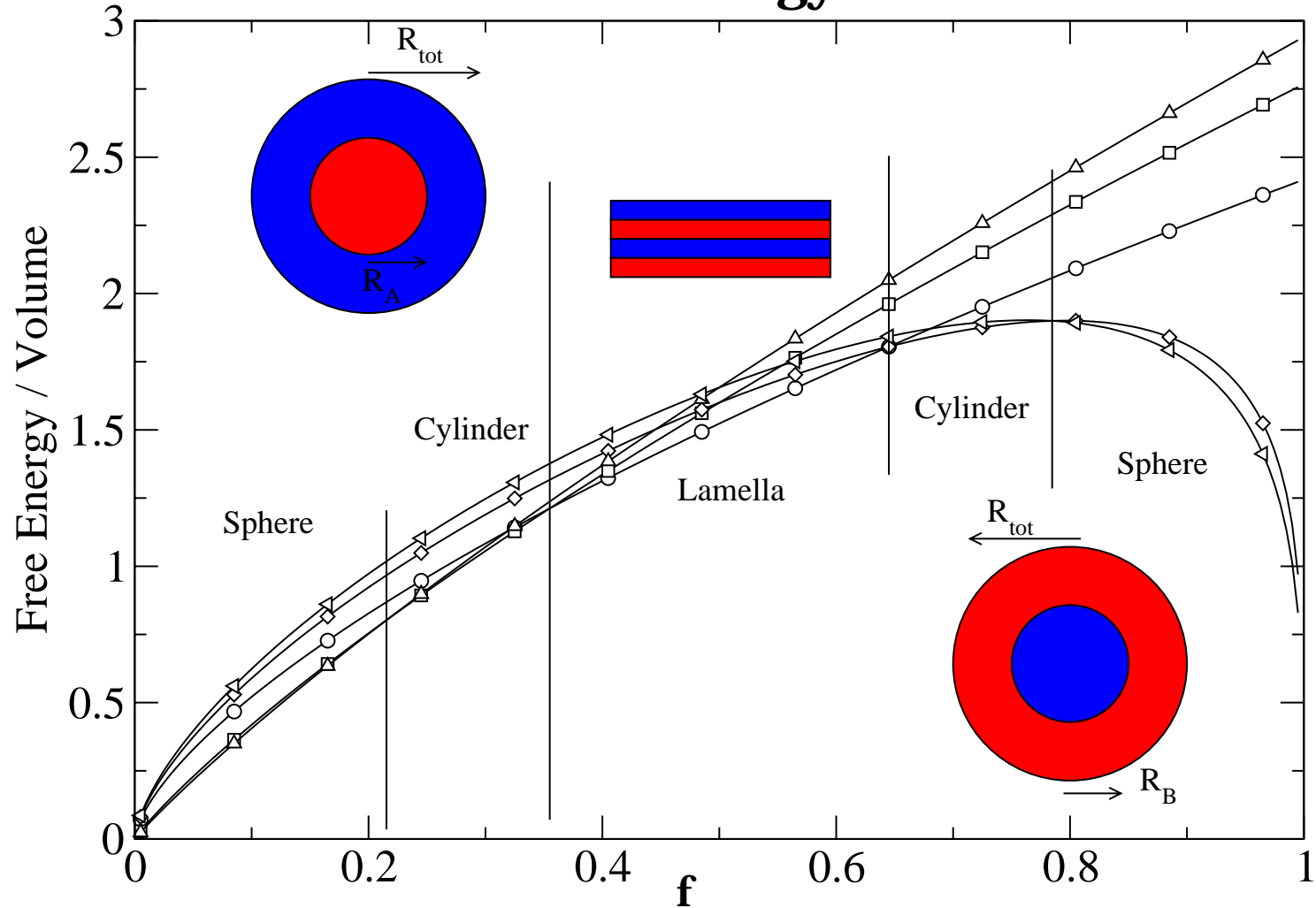


□ **Peak diverges when**

$$(\chi N)_{\text{spin}} = \frac{1}{2f(1-f)} + N \sqrt{\frac{\pi B}{3f(1-f)^3}}, \quad N\sqrt{B} = \text{charge/chain}$$

# ● Strong-Segregation Limit

□ Just a balance of surface energy and electrostatics

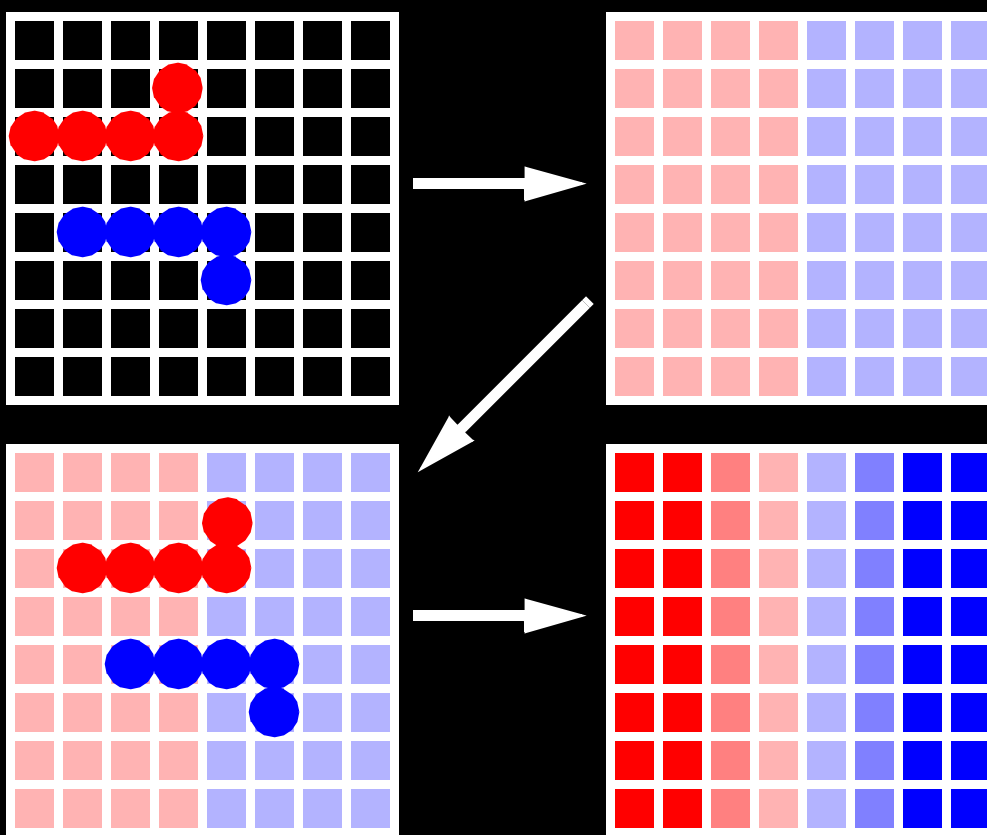




# ● Self-consistent Lattice Model

Fleer, Cohen, Scheutjens, Cosgrove, Vincent, *Polymers at Interfaces* Chapman and Hall, London 1993

## □ Lattice model



1. Start with empty lattice
2. Throw down polymers at random
3. Calculate average monomer densities
4. Regrow the chains
5. Recalculate the monomer densities
6. Repeat

## □ Charged blend, lattice electrostatics.

## ● **Lattice Electrostatics**

---

### □ **Discretize Laplacian:**

$$\nabla^2 \varphi \Rightarrow \varphi(\mathbf{x}, y + 1) + \varphi(\mathbf{x}, y - 1) + \varphi(\mathbf{x} + 1, y) + \varphi(\mathbf{x} - 1, y) - 4\varphi(\mathbf{x}, y)$$

### □ **Gauss' Law discretized:**

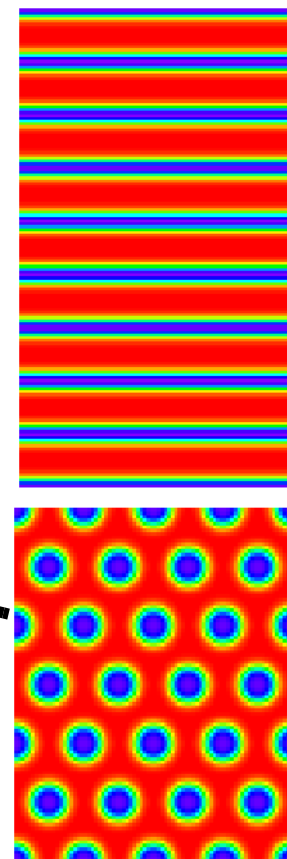
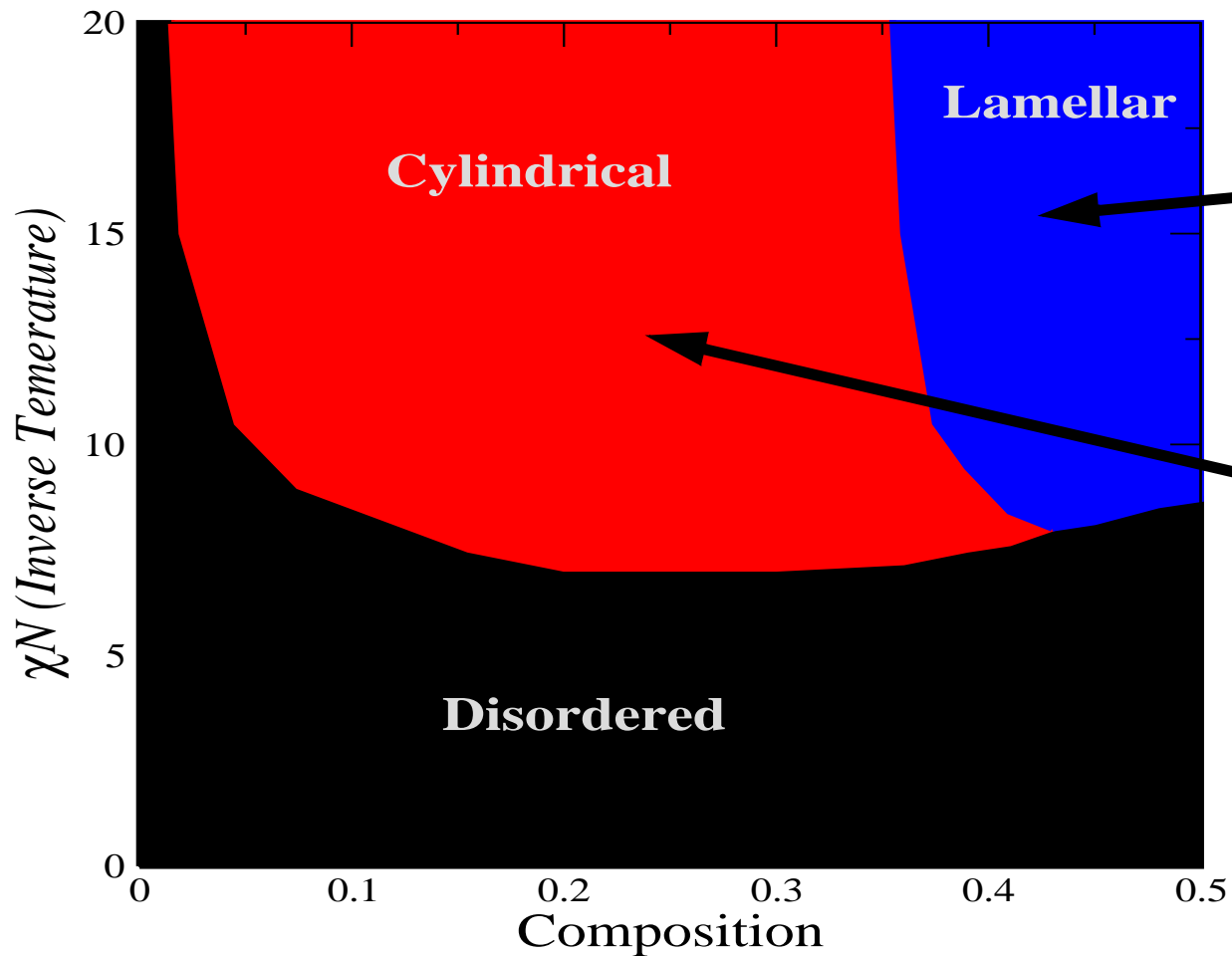
$$\nabla^2 \Phi = 4\pi(\rho_A \varphi_A + \rho_B \varphi_B)$$

### □ **Solve for $\Phi$ , electrostatic potential, involves inverting a linear operator on the lattice**

### □ **Solved numerically at each iteration by direct inversion.**

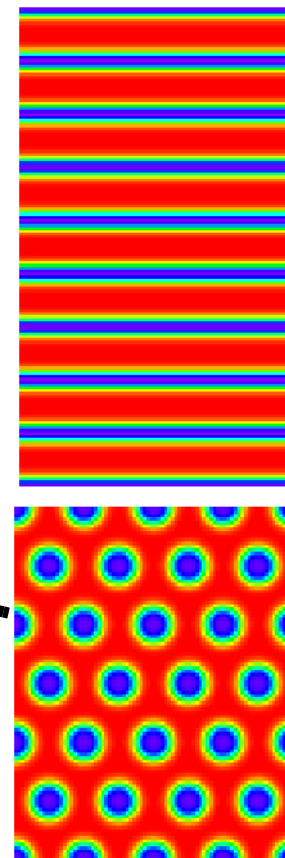
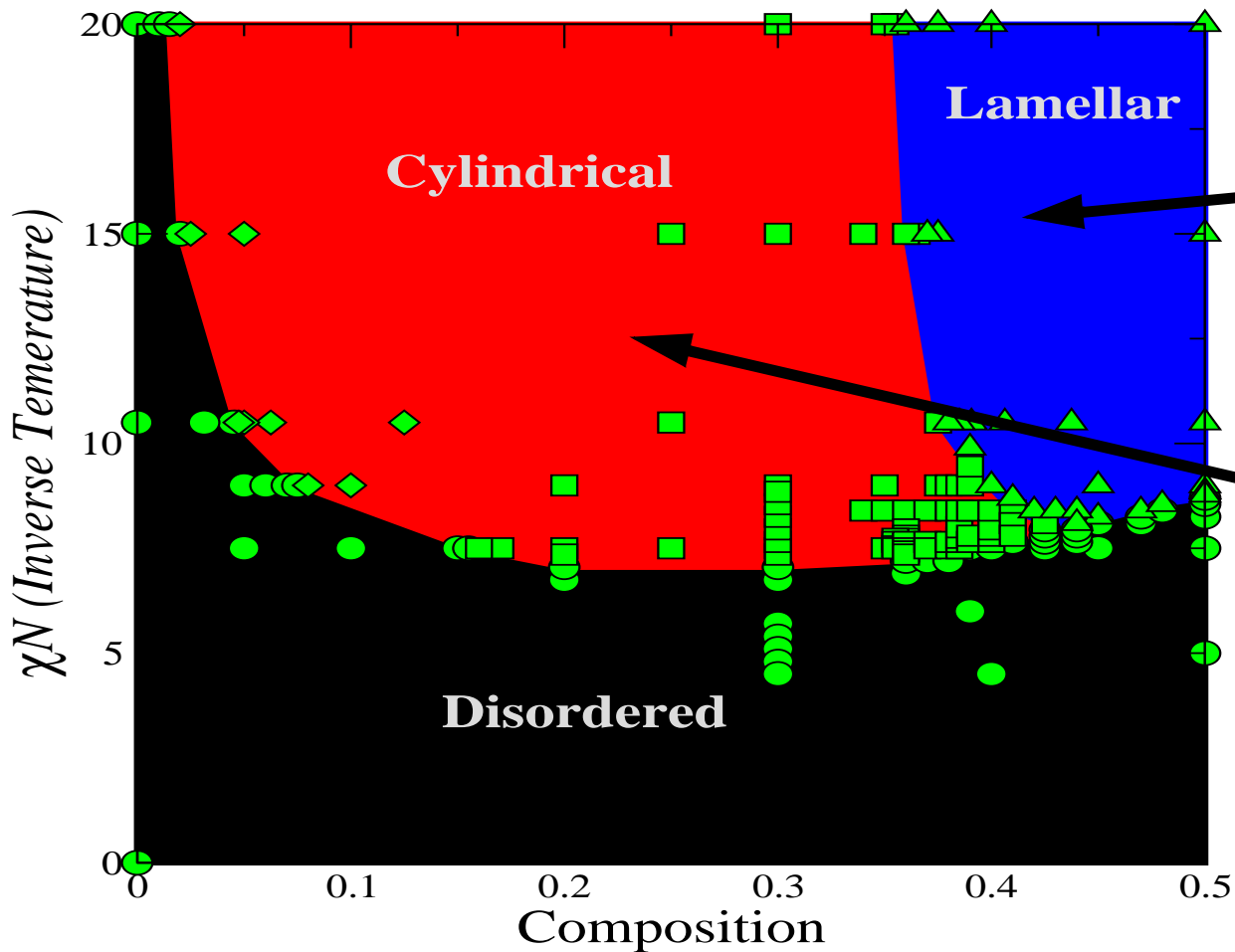
● **Microphases (just like block copolymers)**

Charge/monomer=0.01 N=150

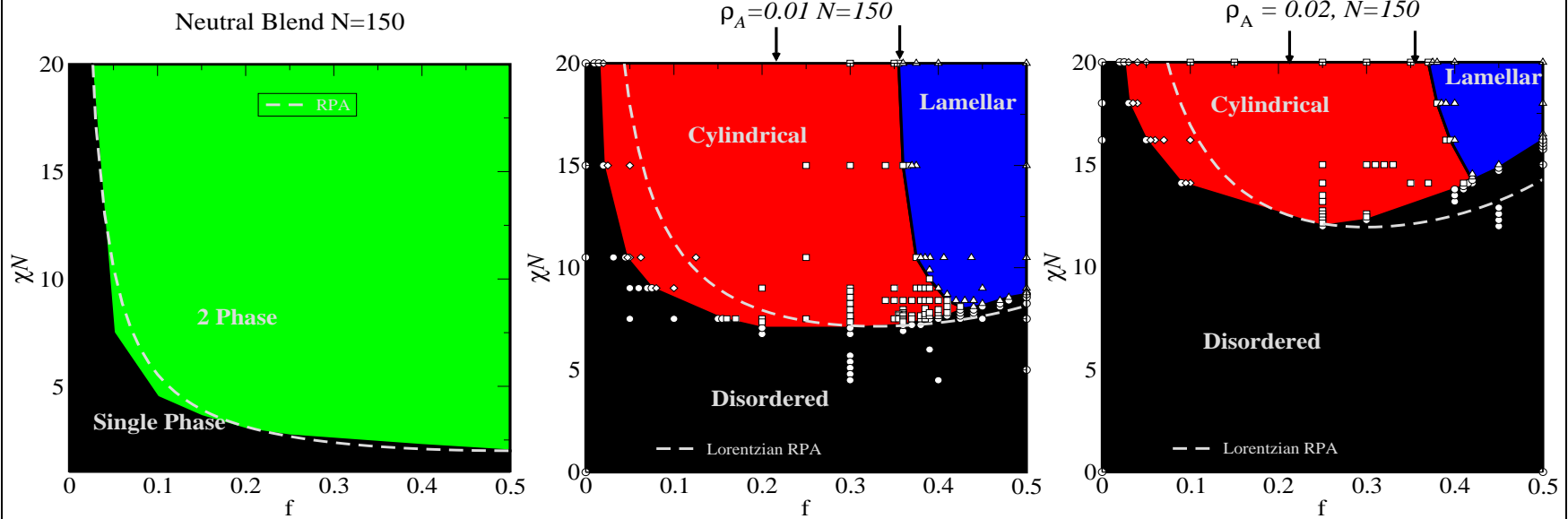


● **Microphases (just like block copolymers)**

Charge/monomer=0.01 N=150



# ● Charge compatibilizes the blend

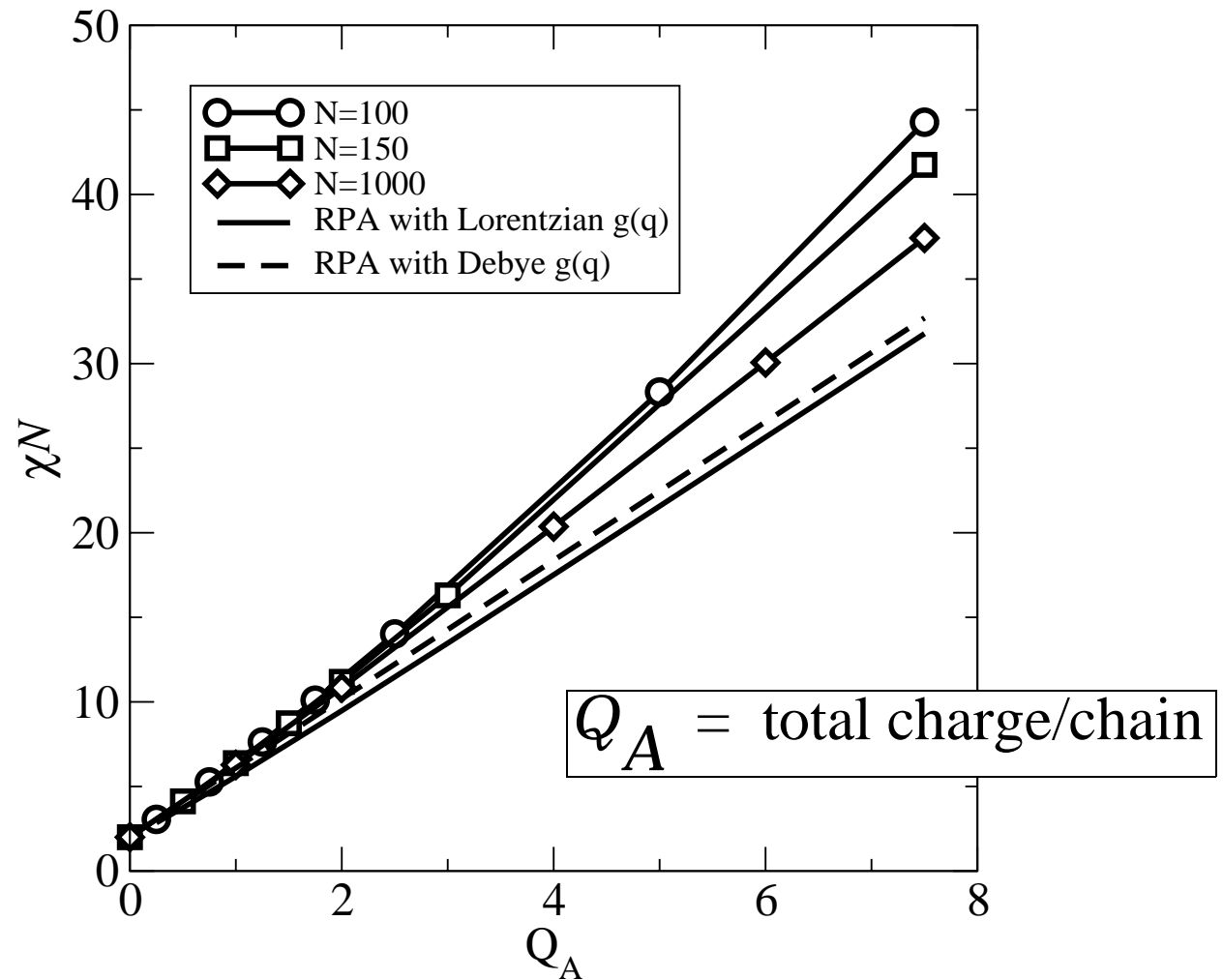


Increasing charge  $\longrightarrow$

- Simple architectures (just homopolymers) but complex patterns.
- Long-range vs. short-range, generic physics.

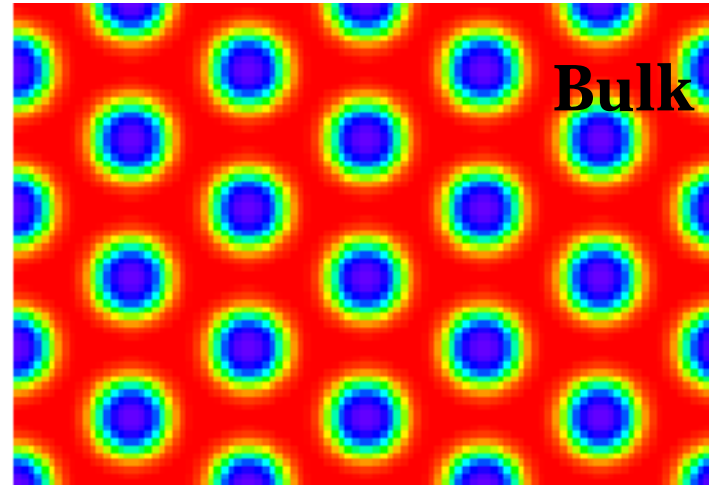
## ● Comparison, SCF RPA

### □ Disordered-Lamellar transition for $f = 1/2$ :



# ● Films

- Lower surface held at a constant potential
- Upper surface is vacuum
- Confinement and external field controls morphology

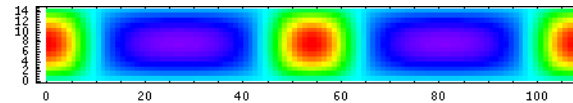
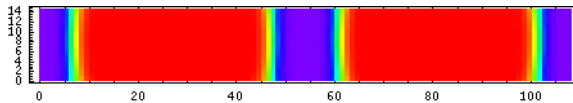


**Film**

$\Phi$

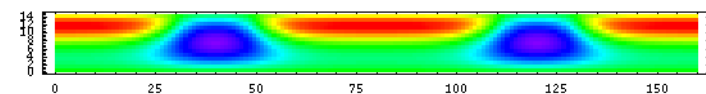
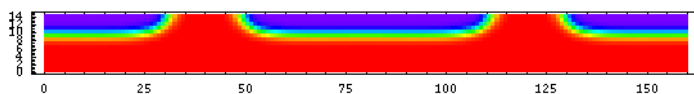
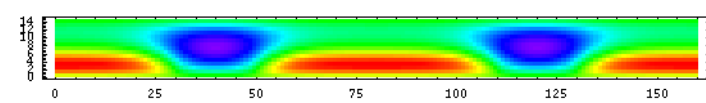
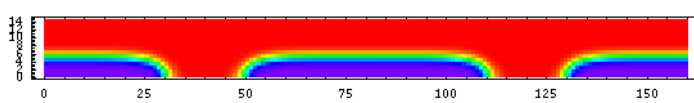
vacuum

**No  
Field**



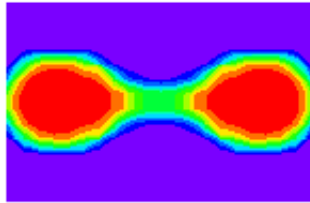
**grounded**

**External  
Field**

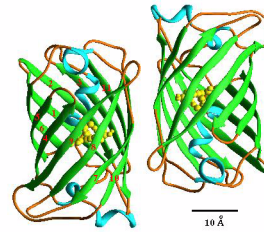


## ● Conclusions

- Single-molecule “beads” generic folding problem



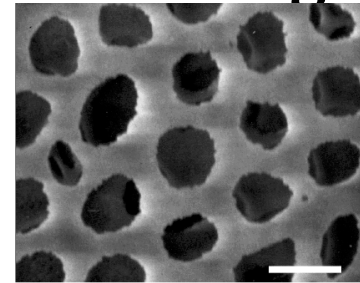
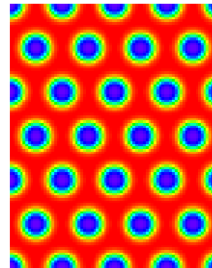
Other foldings



<http://www.bioc.rice.edu/Bioch/Phillips/gfpbio.html>

- Charged blends make microphases Length scale controlled by charge/chain not molecular weight

→ Photonic crystals?



Imhof and D.J. Pine, *Advanced Materials* 10, 697-700 (1998).

- Charge-separated layers, Polymer LED's

